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Tuning main-group s-block metal Mg as a promising single-atom electrocatalyst for N₂ fixation: A DFT study

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Abstract: The electrocatalytic nitrogen reduction reaction (NRR) can transform nitrogen and protons from aqueous electrolytes to ammonia by using renewable electricity under ambient conditions, which is a promising technology to replace the Haber-Bosch process. However, this technology is extremely challenging as it requires highly active electrocatalysts to break the stable triple-bonds of N₂. With p bands, main-group s-block metals have been rarely explored in NRR compared with transition metals. Herein, we employ first-principles calculations to propose a Mg single atom catalyst as a promising high-performance electrocatalyst for NRR, where Mg atom is coordinated with four oxygen atoms within graphene (Mg-O₄). Our results reveal that N₂ can be efficiently activated on Mg-O₄ and reduced into NH₃ through the alternating mechanism. Moreover, ab initio molecular dynamics simulations demonstrate the Mg-O₄ structure has high stability.

Keywords: electrocatalytic N₂ reduction reaction; main-group metal; single atom catalyst; firstprinciples calculations

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1 Introduction

Ammonia (NH₃) is an essential headstone in global industrial chemicals as it is a fundamental raw feed to produce nitric acid, ammonium salt, and chemical fertilizers^[1-3]. Currently, more than 90% of NH_2 is produced by the Haber-Bosch process^[4]. However, this process requires high pressure (15 - 40 MPa) and temperature (350-550 °C), which consumes 1% - 2%of the world's annual energy $supply^{[5-8]}$. Therefore, more attention has been paid to a greener and simpler way to product NH₂. A promising way for the sustainable production of NH₃ under thrambient condition is the electrocatalytic N₂ reduction reaction (NRR), which could provide proton source via electrolysis of water molecules by renewable energy technology^[9-12]. However, the electrocatalytic NRR, involving cleavage of the strong dipole moment of the $N \equiv N$ triple bond, is extremely challenging ^[13, 14]. Therefore, exploring highly active electrocatalysts for

NRR is highly desirable.

Benefiting from the occupied and unoccupied d orbitals, transition metals have been widely applied for the electrocatalytic NRR. For example, a Fe-doped W₁₈O₄₉ nanowire had been reported for the electrocatalytic NRR, which exhibited a high NH₂ Faradaic efficiency of 20.0% at an overpotential of -0.15 V versus reversible hydrogen electrode $(RHE)^{[15]}$. In another case, a S-rich MoS₂ nanosheet achieved a high ammonia production yield rate of 43.4 µg per h per mg catalyst^[16]. In contrast, main-group metals in sblock have been rarely explored in the electrocatalytic NRR. Nonetheless, in nature, metalloenzymes containing s-block main-group metals can catalyze many challenging biological reactions^[17]. For example, chlorophyll, consisting of a porphyrin ring with a central atom of magnesium, can transform CO₂ and H₂O to carbohydrates by capturing the sunlight^[18]. Calcium cofactor plays an important role in the ferroportin transport activity^[19]. However, the poor electrical

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conductivity of metalloenzymes limits their application in the electrocatalytic NRR. Anchoring s-block maingroup metal atoms onto an appropriate substrate with good conductivity may be a solution.

Graphene, consisting of a single layer of sp^2 hybridized carbon, has high electric conductivity and stability, which is an ideal support to anchor metal atoms^[20, 21]. For example, Yao et al. prepared atomically dispersed Ni on the defective graphene^[22]. Monodisperse cobalt atoms embedded in nitrogen-doped graphene was successfully synthesized by Ji's group^[23]. It is worth noting that the nitrogen atoms in</sup> catalyst might disturb its catalytic performance for NH₂ synthesis by electrocatalysis method. Hu and co-workers reported that tetragonal Mo₂N would undergoes decomposition and result in the generation of ammonia, instead of catalyzing electrochemical reduction of $N_2^{[24]}$. Besides, synthetic steps for introducing nitrogen into graphene including calcination in NH₃ atmosphere or the use of compounds containing NH₂ components such as melamine and urea will also result in the false performance of catalysts^[25, 26].

Motivated by these considerations, herein, we perform first-principles calculations to explore the potential of s-block main-group element Mg atom decorated by the defective graphene substrate for NRR under ambient conditions. Our results reveal that the structure of Mg atom anchored on the graphene substrate coordinating with four O atoms $(Mg-O_4)$ has a strong activation effect on nitrogen molecules, which could efficiently reduce N_2 into NH_3 by the alternating mechanism. Furthermore, the result of ab initio molecular dynamics simulations also demonstrated the

thermal stability of Mg-O₄ at the room temperature.

2 Computational details

In this work, we implemented the Vienna Ab initio Simulation Package (VASP) to conduct all calculations through spin-polarized density functional theory (DFT)^[27]. The exchange-correlation energy functional was described in the Perdew-Burke-Ernzerhof (PBE) generalized gradient approximation (GGA) with projector augmented wave (PAW) pseudopotentials^[28, 29]. The cut-off energy of 400 eV were employed. The convergence threshold of energy and force were 1×10^{-5} eV and 0. 02 eV \cdot Å⁻¹. respectively. The k-point grids are $5 \times 5 \times 1$ for geometry optimization. In order to minimize the periodic interaction, a vacuum layer of 20 Å along the vertical direction was set. Chemical-bonding analyses were carried out with the computer program LOBSTER (Local Orbitals Basis Suite Towards Electronic-Structure Reconstruction)^[30, 31]. Ab initio molecular dynamics simulations (AIMD) were employed to evaluate the thermodynamic. The time step is set as 2 fs and the simulation is performed using the canonical ensemble (NVT) in the temperature of 300 K with an optimized initial structure.

The binding energy (E_b) between the Mg atom and the O-modified graphene substrate was calculated by the following equation:

$$E_{\rm b} = E_{\rm catalyst} - E_{\rm O/C} - E_{\rm Mg}$$

Furthermore, the Gibbs free energy (ΔG) for each reaction step in different NRR pathways was defined as follows:

$$\Delta G = \Delta E + \Delta E_{\rm ZPE} - T\Delta S + \Delta G_{\rm U} + \Delta G_{\rm pH}$$

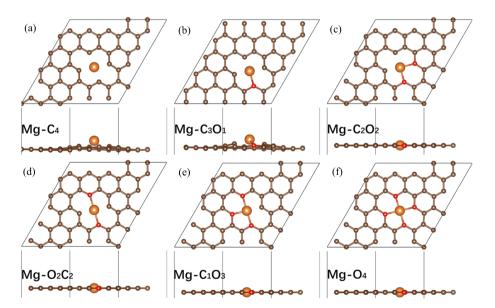


Figure 1. the optimized structure of six models for Mg-O/C. (a) Mg-C₄, (b) Mg-C₃O₁, (c) Mg-C₂O₂, (d) Mg-O₂C₂, (e) Mg-C₃O₁, (f) Mg-O₄.

 ΔE is the electronic energy difference between the product and the reactant species, ΔS and ΔE_{ZPE} are the changes in entropy and zero-point energy, respectively. ΔG_{U} represents the effect of applied bias and is equal to eU. ΔG_{pH} is the contribution of H⁺ and is equal to $-k_{\text{B}}T \times \ln 10 \times \text{pH}$, where k_{B} is the Boltzmann constant. The value of pH is set as 0 to produce thermodynamic values comparable to those of other works.

3 Results and discussion

3.1 Anchoring Mg atom by defective graphene

Previous studies have demonstrated that divacancy is the most common defect present in 2D graphene and it possesses impressive ability for anchoring transition metal atoms which present effective catalytic activity for some electrocatalysis reactions, such as O₂ reduction reaction (ORR), CO₂ reduction reaction^[32-34]. The different coordination element around metal atom would result in diverse electron structure of metal active site. Thus, we design six models where Mg atom is coordinated with different number of O and C atoms. According to the number of O and C coordination atoms, we named those six models as $Mg-C_4$, Mg- C_3O_1 , Mg- C_2O_2 , Mg- O_2C_2 , Mg- C_3O_1 , Mg- O_4 , respectively. The models of Mg- C_2O_2 and Mg- O_2C_2 represent the different coordination structure with Mg atom. The optimized structures of the six models are shown in Figure 1. There is no obvious distortion or collapse of O-modified graphene substrate. Moreover, when the number of O as coordination atoms is greater than two, the Mg atom is well anchored in the graphene layer as shown in the side-view of models. This indicates that the divacancy defect in graphene is also appropriate for anchoring the main group element Mg. Furthermore, the calculated binding energies range from -0.57 to -4.30eV (Figure S1), demonstrating that they are thermodynamically stable.

3.2 Adsorption of N₂

It's well-known that the chemisorption of N₂ molecule on the surface of catalysts is the prerequisite for an efficient NRR process. The effective chemisorption determines whether a catalyst is able to activate N_2 molecule. Previous studies have shown that side-on and end-on patterns are the two ways of N₂ absorption on the catalyst surface^[35, 36]. We firstly verified both absorption modes of N2 molecule on Mg atom. As shown in Figure S2, side-on absorption pattern changed into end-on mode after structure optimization in the all models. This result is in consensus with M. A. Duncan's research conclusion, which demonstrated the activation of N₂ on Mg atom via the linear configuration rather than the side-on configuration^[37]. Currently, it is widely believed that the activation of N₂ is via an " acceptance-donation " process^[38], this mechanism indicates the mutual transfer of electrons between catalysts and N2 molecule. Figure 2 presents the difference charge density of Mg-C/O with the adsorption of N2, where the charge accumulation and depletion can be clearly observed for both N2 molecule and Mg-C/O catalyst. Mutual charge transfer in other five models indicates N2 molecule is effective adsorbed by Mg atom anchored in the O-modified divacancy defective graphene.

However, we simultaneously noticed that the charge accumulation and depletion phenomenon of N_2 and catalyst in model Mg-O₂C₂ is separated and their

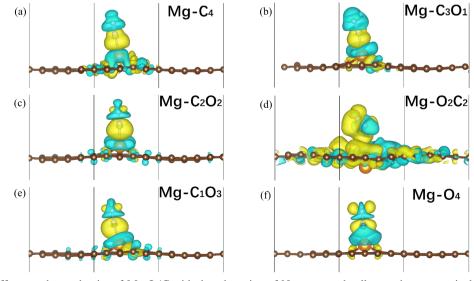


Figure 2. The difference charge density of Mg-O/C with the adsorption of N_2 , cyan and yellow regions respectively represent positive and negative charge regions and the isosurfaces value is 0.006 e/Å³.

don't overlap in space (Figure S3). charges Interestingly, the different site of coordinated O around Mg center in model Mg-O₂C₂ and Mg-C₂O₂ leads to distinct ability for N2 adsorption. In fact, different coordinated structures determine the symmetry of the electronic structure of the Mg center and further exert influence on electronic orbit overlap between the Mg center and N₂ molecule. Figure 3 shows the projected density of states (pDOS) of the N_{ads} atoms in $\ensuremath{\,^*}N_2$ directly bonded to the Mg atom. Compared Figures 3 (a) to 3(d), the area of the 3s and 3p bands of the Mg atom in model Mg-C₂O₂ overlapped with the 2p orbitals of N_{ads} is obvious larger than that in model Mg-O₂C₂, which indicates that the possibility of electrons exchange between N_{ads} and Mg enhances and stronger binding between N_{ads} and Mg-C₂O₂. Furthermore, the strength of bonding between * N2 and Mg-C/O can be understood by projected crystal orbital Hamilton populations (pCOHP) analysis (Figure 3(a) - (f)). The positive and negative of integrated pCOHP values represent the bonding and anti-bonding respectively. As shown in the right part of Figure 3(a), the integrated area of the model $Mg-O_2C_2$ is very small, which is corresponding to the result of pDOS analysis. The integrated pCOHP values up to the Fermi level for Mg-N_{ads} and N_{ads}-N of six models are listed in Table S1, we

thus could analyze the binding strength of N₂ adsorption in a quantitative way. The more negative of this value. the more anti-bonding state is above the Fermi level and more bonding state is under the Fermi level, which indicates stronger binding between adsorbent and catalyst. As the value for Mg-N_{ads} decreases from model Mg- O_2C_2 to Mg- O_4 , while the value for N_{ads} -N increases, suggesting a stronger bonding between Nads and Mg atoms and a weaker N-N bond. More importantly, as shown in Figure 3 (g) and 3(h), we further notice that the bond length of Mg-N_{ads} and N_{ads}-N are also consequence with this trend. It is well-known that the change of N-N bond length determines the degree of nitrogen activation. Therefore, the model Mg-O4 possesses the optimum ability for N2 activation in above-mentioned six models. Simultaneously, we observe that the Bader charge of adsorbed N2 molecule present the same tendency with the N-N bond length. This result provides full proof for that the activation mechanism of Mg-C/O structure coincides with the π back-donation process, which weakens the N-N bond and strengthen the metal-nitrogen bond at the same time.

In addition, we further explore the HER activity of $Mg-O_4$ structure. The optimized structure of the adsorbed H on $Mg-O_4$ is shown in Figure S4. We can

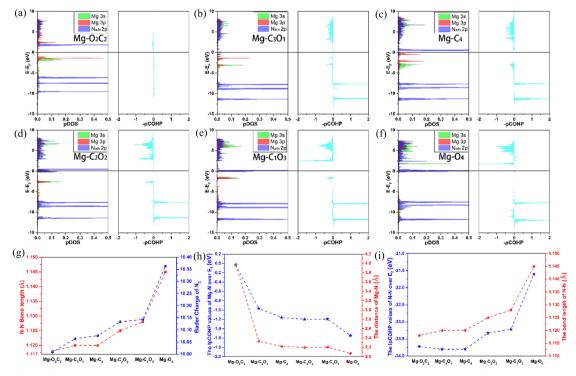


Figure 3. (a)–(f) Projected density of states (pDOS) and projected crystal orbital Hamilton populations (pCOHP) for the N_{ads} atoms of * N_2 directly bonded to the metal atoms of the six models. The relationship between the integrated pCOHP values up to the Fermi level and the bond length of (g) Mg-N, (h) N-N. (i) The relationship between the Bader charge and bond length of adsorbed N_2 molecule.

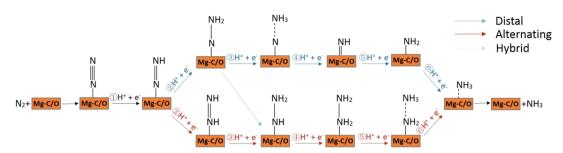


Figure 4. Schematic depiction of the three reaction mechanisms for the electrochemical NRR on Mg-C/O catalyst.

see the single H can also adsorbed by Mg atom. The Gibbs free energy diagrams indicate that H atom is more feasible adsorbed by Mg atom compared to N_2 molecule. However, due to the strong adsorption energy, the H is difficult to desorb from Mg atom to form H₂. In the other words, once the N₂ molecule is adsorbed on the catalyst, the NRR is prior to occur. Thus, it is suggested that the selectivity of NRR would be higher than HER if less H atom in the electrocatalysis environment.

3.3 NRR catalytic mechanism on Mg-C/O

Typically, there are three possible reaction pathways for the N₂ end-on adsorption configuration on the Mg-O/C catalysts surface, i. e., distal, alternating, and hybrid pathways^[39, 40]. As shown in Figure 4, in the mechanism of the distal pathways, the proton and electron pairs $(H^+ + e^-)$ are prone to interact with the terminal N atom at first, which results in the breakage of the N - N bond and the release of the first NH_3 molecule. After that, it sequentially attacks the other N atom to produce the second NH_3 . In comparison, the hydrogenation process of two N atoms would occur successively in the alternating pathways. In the hybrid pathway, the first two steps are the same as those in the distal pathway, and the following steps are along the alternating pathway. Given that the adsorption strength for N₂ of catalysts could directly affect the NRR catalytic performance, we firstly explored those three reaction pathways on the Mg-O₄ structure. In the Figure S5 and S6 are the top-view and side-view of optimized configurations along distal and alternating reaction pathways, respectively. We can observe that the structure of Mg-O4 don't collapse or distortion after the adsorption of various N_xH_y intermediates. Here, we did not list the hybrid pathway because the intermediate adsorbents of hybrid pathway is consistent with the other two ways.

As illustrated in Figure 5, the Gibbs free energy diagrams of above-mentioned three pathways on $Mg-O_4$ show both uphill and downhill steps, which means endothermic and exothermic steps coexist during the NRR process. Generally, the potential-determining step

(PDS) of a reaction is the step with the maximum Gibbs free energy change (ΔG_{I}) of endothermic step. Furthermore, the corresponding applied potential with $\Delta G_{\rm I}$ to make all steps of reaction exothermic is named as $U_{\rm I}$. The Gibbs free energy change $(\Delta G_{\rm I})$ is a very important indicator to evaluate the catalytic activity in thermodynamic perspective. In Figure 5(a), we can find that the first hydrogenation of the adsorbed $N_2(*N_2)$ + H⁺+ e⁻ \rightarrow * NNH) is endothermic with a free energy uphill of 1.04 eV. The similar phenomenon is also observed in the previous investigation of transition metal and main-group element based catalyst, such as Fe-N- $C^{[36]}$ and B-rich COFs^[41]. In the following reaction procedures, the distal pathway undergoes another endothermic step with 1. 35 eV Gibbs free energy (Figure 5(b)), which releases the first NH₃ molecule. In comparison, the Gibbs free energy of endothermic steps in alternating and hybrid pathways don't overpass the first hydrogenation step. Thus, the potential-limiting step of distal mechanism is the third hydrogenation step with a NH₃ molecule release, while the first hydrogenation steps of alternating and hybrid pathways are their respective PDS. As the Gibbs free energy change $(\Delta G_{\rm I})$ of the potential-determining step in distal mechanism is larger than that in alternating and hybrid pathways, the NRR prefers to follow the alternating and hybrid pathways on model Mg-O₄ structure.

After that, we continued to explore those three NRR pathways on other five models. The adsorbed configurations of various N_xH_y intermediates on other models are illustrated in Figures S7–S16, meanwhile the corresponding Gibbs free energy diagrams are presented in Figure S16. We notice that all the distal pathways of the six models present three uphill steps: the first hydrogenation of N_2 and the release of two NH₃ molecules. Thus, the Gibbs free energy of those steps determines the reaction rate in distal mechanism. Interestingly, Figure 6 (a) shows us two linear relationships: with the number of O atoms increasing, the Gibbs free energy required for the first and second NH₃ molecule release is linear scaling with the Gibbs free energy of the adsorbed

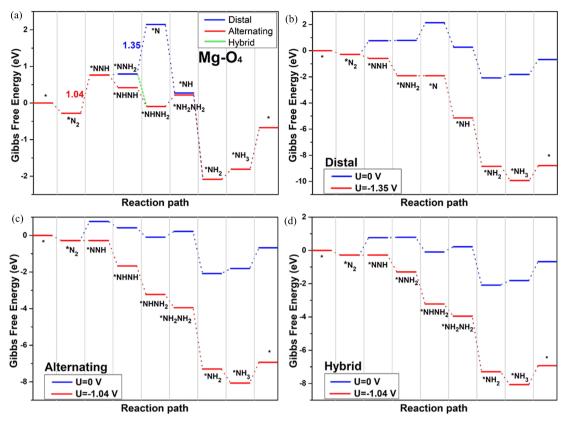


Figure 5. (a) Gibbs free energy diagrams for the NRR on model Mg-O₄ without applied potentials. Comparison of Gibbs free energy diagrams at 0 V (blue) and U_{L} (red) via (b) distal, (c) alternating and (d) hybrid mechanisms.

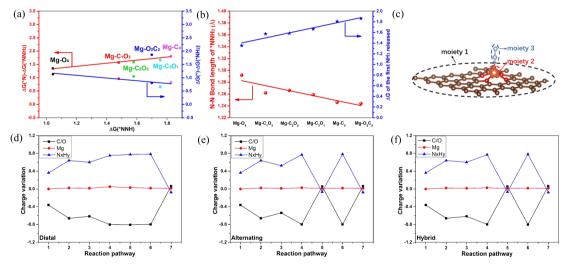


Figure 6. (a) The relationship of three uphill steps through distal pathway, left and right axis are the Gibbs free energy required for the first and second NH₃ molecule release, respectively, *x* axis is the Gibbs free energy of the first hydrogenation of the adsorbed N₂. (b) The trend between N-N bond and ΔG of second NH₃ molecule release. (c) The three moieties of adsorbent structure. Charge variation of each moiety between the present step and the initial structure for Mg-O₄ along the (d) distal, (e) alternating, and (f) hybrid pathway.

 N_2 . However, their linear relationships are in opposite direction. This characteristic is different from the situation of transition metal surfaces, indicating that the Mg-O/C structures break the scaling relation between

the energetics of uphill steps and could be candidate materials for ammonia synthesis^[42]. Meanwhile, the first NH₃ release step requires larger Gibbs free energy (ΔG_1) than the second NH₃ release (ΔG_2) , which means the former step determines the NH₃ yield rate. Furthermore, Figure 6(b) presents that the trend of N-N bond length of adsorbed intermediate NNH, in the distal pathway is inverse with the tendency of ΔG_1 . In other words, the longer of N-N bond length of * NNH₂, the less energy needed for the first NH₂ release. In fact, a longer N-N bond length represents a weaker binding strength, thus energy required for the N-N cleavage decreases. As shown in Figures S18 (a) - (f), in the alternating pathway, the N-N length was elongated to 1.483 Å before the first NH₂ release step on all six Mg-C/O models, while the longest N-N bond length before the first NH₃ release step in the distal pathway is just 1.296 Å of model Mg-O₄. Compared to distal pathway, alternating pathway is more favorable for NRR catalysis on the model Mg-C/O structures. Coincidentally, with the number of O atoms coordinated with Mg increasing, the $\Delta G_{\rm L}$ of different models decreases in alternating pathway. Therefore, the model Mg-O4 possesses the optimized NRR catalytic activity through the alternating mechanism.

To gain further insights into the catalytic effect of the Mg-C/O on the N_2 -fixation, we performed the charge population analysis. As illustrated in Figure 6 (c), each adsorption structure is divided into three moieties: O-modified graphene (moiety 1), Mg atom (moiety 2), and the adsorbed $N_{x}H_{y}$ intermediates (moiety 3). Here, we defined the charge variation as the Bader charge difference between each moiety in the present step and that of the initial Mg-C/O structure without adsorbents. The charge variation at step 1 represents the mutual transferred charges between the three moieties at N₂ adsorption step. Figure 6 (d)-(f)are the charge variation diagrams of the distal, alternating, and hybrid pathways in $Mg-O_4$, respectively. Impressively, the charge variation of Mg atom presents little fluctuation during the whole reaction process of all the three pathways. Meanwhile, the charge variation between O-modified graphene (moiety 1) and the bounded $N_r H_v$ intermediates (moiety 3) is negative correlation. Superficially, the electrons for activating N_xH_y intermediates seem to originate from the O-modified graphene and have nothing to do with the Mg atom. In fact, the O-modified graphene substrate gains 1. 67 electrons from the Mg atom after forming Mg-O₄ structure and the $N_{y}H_{y}$ intermediates get 0.77 electrons at most during the whole NRR process. Therefore, it's reasonable to believe that the valence electrons of Mg atom redistribute between the Omodified graphene and N_xH_y intermediates during the whole reaction procedure. In addition, as shown in Figure S19, the 4O-modified graphene model without the Mg atom anchoring presents no adsorption ability for

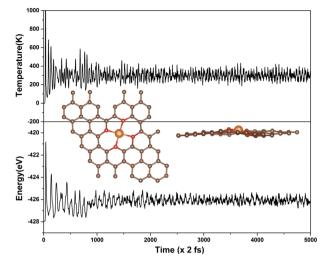


Figure 7. Variations of temperature and energy against the time for AIMD simulations of $Mg-O_4$, insert are top and side views of the snapshot of atomic configuration. The simulation is run for 10 ps with a time step of 2 fs at 300 K.

 N_2 molecule. Thus, the Mg atom is indispensable for NRR reaction in Mg-C/O structure. From above analysis, we can identify the role of each moiety in Mg-O/C structure: the 4O-modified graphene acts as an electron reservoir ascribing to the high charge density and the Mg atom is the active site for N_2 -fixation and the transmitter for the electron transfer, and the collaboration between them contributes to the system to be a novel catalyst for N_2 -fixation.

The thermal stability of the Mg-O₄ structure was evaluated by using AIMD simulations at 300 K. As shown in Figure 7, there is no significant geometrical distortion for Mg-O₄ during 10 ps simulation, indicating the relatively high stability. Due to the high stability of the catalyst, we envision that the Mg-O₄ structure could possess full advantages in terms of N₂ fixation at ambient conditions.

4 Conclusions

In summary, we have designed the main-group s-block metal single atom electrocatalyst, Mg-O₄, with excellent activity for N₂ fixation. The mutual charge transfer between N₂ and Mg-O₄ confirms the "acceptance-donation" process, which is similar to the traditional metals for the capture and activation of N₂. The subsequent N₂ reduction process prefers to occur through the alternating mechanism due to the longer bond between two N atoms before releasing the NH₃ molecule. Moreover, our computed result indicates the number of O coordination atom plays an important role for the catalytic activity of Mg atom. In addition, this catalyst exhibits a high thermal stability under ambient environment, which is suitable for sustainable NH₃ production.

Supplementary data

Supplementary data are available at J. Univ. Sci. Tech. China online.

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Conflict of interest

The authors declare no conflict of interest.

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主族 s 区金属 Mg 电催化 N₂ 还原反应的第一性原理计算研究

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摘要: 电催化 N_2 还原反应(NRR)可以在温和的条件下利用可再生的电力将氮气和质子从电解质水溶液转化 为氮,被认为是一种非常有前景的替代哈伯法合成氨技术. 但是这项技术也面临巨大的挑战,因为 N_2 分子中的 $N \equiv N$ 键非常牢固,需要高活性的催化剂才能将其断裂. 和过渡金属相比,主族 s 区金属在 NRR 中很少被研究. 本文采用第一性原理计算的方法,发现氧掺杂的石墨烯锚定的镁单原子催化剂(Mg-O₄)是一种高活性的 NRR 电催化剂. 理论计算结果表明, N_2 分子能有效地被 Mg-O₄ 活化,并通过非解离交替机制被还原成 NH₃. 此外, 分子动力学模拟结果显示 Mg-O₄ 具有高的稳定性.

关键词: 电催化 N, 还原反应; 主族金属; 单原子催化剂; 第一性原理计算